

# First-principle investigation of structural, electronic and magnetic properties of Co<sub>2</sub>VIn and CoVIn Heusler compounds

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Investigation of the structural, electronic and magnetic properties of full-Heusler  $Co_2VIn$  as well as half-Heusler CoVIn Cobalt based Heusler compounds using density functional theory (DFT) leads to the general conclusion that  $Co_2VIn$  and CoVIn are half-metallic materials with a gap at the Fermi level in the minority states and majority states respectively. A Hubbard-like Coulomb correlation term U has been included in the DFT (DFT+U) for the computation of the electronic and magnetic properties of the compounds. The structural properties have been calculated for the paramagnetic and ferromagnetic phases, and both  $Co_2VIn$  and CoVIn are found to be stable in the ferromagnetic phase. The calculated magnetic moments are 2  $\mu_B$  and 0.9  $\mu_B$  per formula unit for  $Co_2VIn$  and CoVIn respectively. © 2017 Author(s). All article content, except where otherwise noted, is licensed under a Creative Commons Attribution (CC BY) license (http://creativecommons.org/licenses/by/4.0/). [http://dx.doi.org/10.1063/1.4973763]

# I. INTRODUCTION

Heusler compounds consist mainly of full Heusler and half-Heusler families with stoichiometry of X<sub>2</sub>YZ and XYZ respectively, where X and Y are transition metal atoms while Z is a main group *sp* element. Most Co<sub>2</sub>YZ Heuslers exhibit half metallic character and half metallic ferromagnetism.<sup>1</sup> They possess desirable properties such as high spin polarizations and magnetic moments as in Co<sub>2</sub>FeSi with the highest reported Curie temperature (T<sub>c</sub>),<sup>1–5</sup> structural similarity to industrial binary semiconductors<sup>3–5</sup> and variable localized Co magnetic moments.<sup>6</sup> Others include a large spin stiffness in interfaces such as Co<sub>2</sub>MnSi/MgO as compared to 3*d* metals,<sup>7</sup> perpendicular magnetic anisotropy (PMA) energy density comparable to Co/Pd,Pt multilayers as in Co<sub>2</sub>FeAl/MgO<sup>8</sup> and tunable physical properties by alloying with a fourth element as in CoCrFeAl and CoFeAlSi<sup>7</sup> among other intriguing features.

Due to their exceptional electronic structure and outstanding properties, they have found application in Magnetic Tunneling Junctions such as  $\text{Co}_2\text{MnSi/MgO/Co}_2\text{MnSi}$  with Tunnel Magnetoresistance (TMR) ratios of 1900%, surrent-perpendicular-to-plane giant magnetoresistance (CPP-GMR) read heads, spin torque oscillators (STO), spin transistors, surrent magnetic sensors and non volatile magnetic random access memories. Traditionally, ferromagnetic 3d metals have been used in such devices. However, they have a spin polarization of 40% - 50% and they cause problems due to a large difference between their resistance and that of semiconductor substrates.  $^{15-17}$ 



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The magnetic moments of the  $Co_2YZ$  half-metallic ferromagnets follow the localized part of the Slater-Pauling curve. The sp element at the Z site plays an important role for the formation of the magnetic moments at the X site  $^{18}$  and the inclusion of a Hubbard-like Coulomb correlation term U in calculations, respects the partial localization of the d electrons in the transition metal atoms. The local density approximation (LDA) in density functional theory (DFT) and U, (LDA+U) was used to reproduce the measured magnetic moment of 6  $\mu$ B for  $Co_2FeSi$ . It was employed to describe the electronic and magnetic properties of half Heusler CoFeIn and full Heusler  $Co_2FeIn$  compounds and the experimental magnetic state of  $Co_2MnSi$  and  $Co_2FeSi$  was well described using U values that correspond to those for the Coulomb interaction  $U_{dd}$  between d electrons in elemental 3d transition metals, determined prior to the introduction of the LDA+U method.  $^{21}$ 

Heusler compounds  $Co_2VAl$  and  $Co_2VGa$  have been predicted to be half metallic.<sup>22</sup> The next in the series is  $Co_2VIn$ . We have investigated the structural, electronic and magnetic properties of full Heusler  $Co_2VIn$  and half Heusler CoVIn compounds using DFT and DFT+U methods. There are no other experimental or theoretical published results for comparison with the present calculations. Hence, these results can serve as the reference data for further works in this field.

#### II. COMPUTATIONAL DETAILS

We performed DFT calculations based on the Kohn-Sham formalism of spin-polarized density functional theory using the Vienna Ab Initio Simulation Package (VASP) software. It uses a plane wave basis set and projector-augmented wave (PAW) based pseudo-potentials. Structural optimization has been performed using the generalized gradient approximation (GGA) employing the Perdew-Burke-Ernzerhof (PBE) exchange-correlation functional (PBE-GGA). For the determination of the electronic and magnetic properties, the local spin density approximation L(S)DA, GGA and PBE-GGA as well as DFT+U method have been used. Atomic cores were represented by the projector augmented wave method. A Monkhorst-Pack uniform K point grid with 21 x 21 x 21 k points was chosen for geometry optimization and static total energy calculations. Geometries were optimized by relaxing both the unit cell and the positions of all the atoms within the unit cell using the conjugategradient algorithm, until a stopping criterion of energy change less than 10<sup>-6</sup> eV was attained. Uniform cut-off energy of 430eV was chosen for all calculations. The integration over the irreducible part of the Brillouin zone was done using the linear tetrahedron method with Blöchl corrections. U values of  $U_{Co} = 1.92$  eV and  $U_{V} = 1.34$  eV for Cobalt and Vanadium respectively and J = 0.5eV with L(S)DA were applied for the method based on simplified rotationally invariant Dudarev approach implemented in VASP where  $U_{\text{eff}} = U$  - J. U values of 0.8eV and J = 0.5 eV with PBE-GGA and U= 1eV and J = 0.5 eV with GGA resulted in a similar prediction as with L(S)DA + U.

#### III. RESULTS AND DISCUSSION

In this section, we report the structural, electronic and magnetic properties of Co<sub>2</sub>VIn and CoVIn, and explain the latter two in terms of the orbital nature of the wave functions near the Fermi level.

## A. Geometric properties

The optimized lattice constant for  $Co_2VIn$  is 6.001 Å, while that of CoVIn is 5.849 Å as predicted by GGA, while that predicted by LDA is lower as expected at 5.85 Å and 5.7Å for  $Co_2VIn$  and CoVIn respectively. The two Co atoms occupy the 8c ( $\frac{1}{4}$ ,  $\frac{1}{4}$ ) position and V and In occupy the ( $\frac{1}{2}$ ,  $\frac{1}{2}$ ) and (0, 0, 0) positions respectively in the optimized structure of  $Co_2VIn$  while the Co atom occupies the ( $\frac{1}{4}$ ,  $\frac{1}{4}$ ,  $\frac{1}{4}$ ) position and V and In occupy the ( $\frac{1}{2}$ ,  $\frac{1}{2}$ ) and (0, 0, 0) positions respectively in the optimized structure of CoVIn. On relaxation of unit cell and ionic positions, the cubic structure is maintained. Additionally, the  $L2_1$  structure is found to be more stable than the  $Hg_2CuTi$  structure adopted by some Heusler alloys.

The structural properties are calculated in both the paramagnetic and ferromagnetic phases. The total energies optimized in the ferromagnetic phase are lower than the ones in the paramagnetic one, which confirms that the alloys are stable in the ferromagnetic phase. The energy difference between the two phases is 0.876 meV and 0.160 meV for  $Co_2VIn$  and CoVIn respectively.

# **B.** Electronic properties

Spin polarized calculations of  $Co_2VIn$  and CoVIn Heuslers have been carried out at the equilibrium lattice parameters by employing the L(S)DA, GGA and PBE-GGA approximations. The electronic structure of  $Co_2VIn$  obtained from the L(S)DA calculation revealed a gap in the minority states, below the Fermi level resulting in a highly spin polarized band structure at the Fermi level. However, the Fermi level fell within the gap in the minority states when the GGA approximation was employed and the gap widened even further with the PBE-GGA approximation, with the Fermi level moving closer to the center of the gap as shown in Fig. 1.

The electron spin-polarization at the Fermi level is defined by the following expression

$$P = \frac{\rho \uparrow (E_F) - \rho \downarrow (E_F)}{\rho \uparrow (E_F) + \rho \downarrow (E_F)}$$

Where,  $\rho \uparrow (E_F)$  and  $\rho \downarrow (E_F)$  are the spin dependent densities of states at  $E_F$  for the majority and minority-spin cases, respectively. The highest spin polarization is attained from the PBE-GGA calculation. The obtained spin polarizations for all approximations are summarized in Table I.

To obtain a clearer picture of the electronic band structure and to account for electron correlation, we carried out DFT+U calculations in order to treat the electron-electron correlations on localized d states of Co and V. The  $U_{\rm eff} = U - J$ , was used to correct the double counted terms. With L(S)DA approximation, the U values used by Kandpal et.  $al^{22}$  in studying the electronic properties of systems such as Co<sub>2</sub>VAl which is isoelectronic and isostructural to Co<sub>2</sub>VIn, worked excellently predicting 100% spin polarization. In order to reproduce a similar result as with L(S) DA, coulomb correlation values U required for GGA and PBE-GGA were much lower than those with L(S) DA, possibly due to improved estimation of coulomb correlation in these approximations and was determined using the U ramping up method.

The inclusion of U in the calculation causes the band gap in the minority states of  $Co_2VIn$  to open up to an average of 0.2 eV as recorded in Table I, with the lowest conduction band at the G point shifting to higher energies. In addition, 100% spin polarization is realized with all approximations

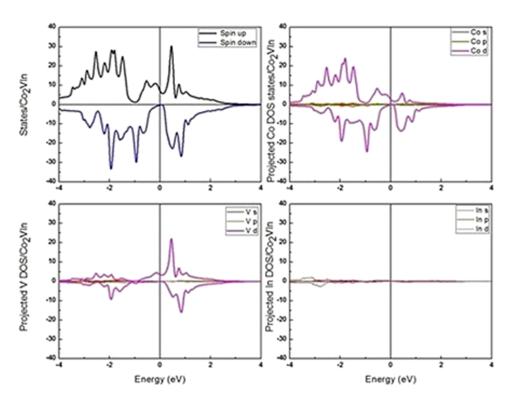


FIG. 1. TDOS and Partial DOS for Co<sub>2</sub>VIn for Co, V and In using PBE-GGA approximation.

TABLE I. The calculated results for spin Polarization and Band Gaps for Co<sub>2</sub>VIn and CoVIn obtained using L(S)DA, L(S)DA+U, GGA, GGA+U, PBE-GGA and PBE-GGA+U approximations.

		$Co_2VIn$	CoVIn		
	Spin Polarization	Minority states Band Gap (eV)	Spin Polarization	Majority states Band Gap (eV)	
L(S)DA	86.28%	0.00	87.32%	0.20	
GGA	98.38%	0.07	88.65%	0.20	
PBE-GGA	99.40%	0.09	94.07%	0.27	
PBE-GGA+U	100.00%	0.19	100.00%	0.08	
L(S)DA+U	100.00%	0.21	100.00 %	0.18	
GGA+U	100.00%	0.20	100.00%	0.18	

with the Fermi level almost at the centre of the gap. This confirms the half metallic characteristic with the charge transport being dominated by the spin-up electrons. The corresponding spin resolved band structures along the high symmetry directions are consistent with the total density of states plots as depicted by the minority spin band structure plot for Co<sub>2</sub>VIn in Fig. 2.

The band gap in  $Co_2VIn$  can be explained to be as a result of hybridization between the two Co and V 3d orbitals. The  $e_g$  and  $t_{2g}$  states of the Co and V sites dominate the part of the plots around  $E_F$  as shown in Fig.1 for both spin channels. A closer look at the atomic resolved DOS in Fig. 1 reveals that V atoms present a broad spin down gap unlike Co sites around  $E_F$ .

The minority states around the gap therefore are localized at the Co sites and do not couple to V. This therefore means that the gap occurs between antibonding states formed after the hybridization between Co 3d orbitals commonly referred to as  $t_{lu}$  ( $d_{xy}$ ,  $d_{yz}$  and  $d_{zx}$ ) represented by the peak below the Fermi level and  $e_u$  ( $d_{z2}$  and  $d_{x2-y2}$ ) states represented by the peak above the Fermi level, that do not hybridize with V states, as they do not transform with the same representation. This therefore explains why the band gaps are small, which is characteristic of d-d band gaps.

The L(S)DA calculation for CoVIn yielded a highly spin polarized total density of states at the Fermi level, with a gap in the majority states and the Fermi level located closer to the valence band. The spin polarization was not 100% as some majority states were present at the Fermi level. The density of states plots were similar for the three approximations the only difference being the slight shift of the Fermi level towards the center of the gap as the approximations were varied from L(S)DA to PBE-GGA resulting in higher spin polarizations. The total and partial density of states plot for CoVIn are depicted in Fig. 3.

The spin resolved band structures obtained using GGA+U for CoVIn revealed that the minority states are conducting while a band gap exists at the Fermi level for the majority states. The GGA+U

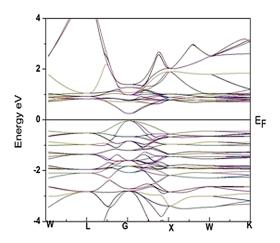


FIG. 2. Spin down electronic band structure of Co<sub>2</sub>VIn at the equilibrium lattice parameter using DFT+U approximation.

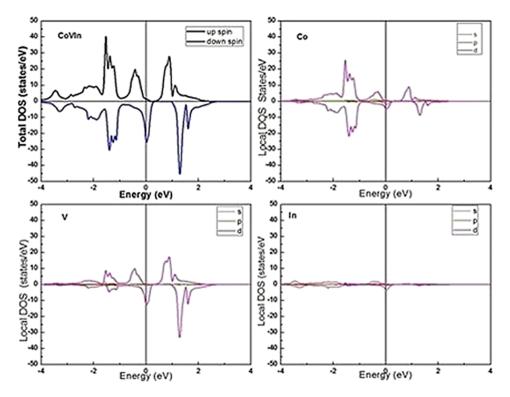


FIG. 3. TDOS and Partial DOS for CoVIn for Co, V and In using PBE-GGA approximation.

scheme resulted in 100% spin polarization of states, with only the minority states at the Fermi level. The partial density of states plots revealed the domination of the 3d electrons of both Co and V around the Fermi level resulting in a hybridization gap with the bonding  $t_{2g}$  ( $d_{xy}$ ,  $d_{yz}$  and  $d_{zx}$ ) states forming the peak just below the  $E_F$  and the peak just above  $E_F$  resulting from the antibonding  $e_g$  ( $d_z^2$  and  $d_x^2$ - $_y^2$ ) states. This is well depicted in Fig. 3. The spin polarization results as well as band gaps are summarized in Table I.

For both Heusler compounds  $Co_2VIn$  and CoVIn, our results show that the majority and the minority spin channels of both Heusler alloys do not display a similar distribution, implying spin polarization. DFT+U emerges as a powerful technique of treating the d states of the transition metals in these type of systems, revealing the expected fully spin polarized electronic structures.

## C. Magnetic properties

The Slater Pauling rule predicts the total magnetic moment in Heusler alloys. For the full Heusler alloy it is  $M_T = Z_T - 24$  and for the half Heusler alloys it is  $M_T = Z_T - 18$ , where  $M_T$  is the total magnetic moment per unit cell and  $Z_T$  is the total number of valence electrons.<sup>23</sup> For Co<sub>2</sub>VIn and CoVIn, the magnetic moments predicted by the Slater Pauling rule are 2  $\mu$ B and 1  $\mu$ B respectively. This comes about when the number of occupied states is fixed at twelve for Co<sub>2</sub>VIn and nine for CoVIn for the minority and majority states respectively due to hybridization resulting in the formation of a half metallic gap typical of Heuslers. In this case, the calculated values are very close to these predicted integer values differing by between 0.024 $\mu$ B and 0.16 $\mu$ B as predicted by L(S)DA and PBE-GGA+U respectively but in full agreement with the GGA moments. These are summarized in Table II together with the moments predicted for each site.

The results for the magnetic moments of  $Co_2VIn$  in Table II follow the exact trend as those obtained from a study of the magnetic properties of  $Co_2VAl$  and  $Co_2VGa$  which are isoelectronic and isostructural to  $Co_2VIn$ , in which Co and V were found to couple ferromagnetically with magnetic moment values of 0.94 and 0.22 $\mu$ B for Co and V respectively for  $Co_2VAl$  and 0.98 and 0.15 $\mu$ B for Co and V respectively for  $Co_2VGa$  using full potential linearised augmented plane wave method

$TABLE\ II.\ The\ calculated\ results\ for\ magnetic\ moments\ for\ Co_2VIn\ and\ CoVIn\ obtained\ using\ L(S)DA,\ L(S)DA+U,\ GGA,$	
GGA+II PRE-GGA and PRE-GGA+II approximations	

		Total Magnetic Moment	Co	V	In
L(S)DA	Co <sub>2</sub> VIn	1.985	0.892	0.235	-0.034
	CoVIn	0.828	0.159	0.596	0.073
L(S)DA+U	Co <sub>2</sub> VIn	2.016	1.037	0.005	-0.063
	CoVIn	0.892	-0.240	1.065	0.074
GGA	Co <sub>2</sub> VIn	2.001	0.928	0.192	-0.047
	CoVIn	0.852	0.095	0.684	0.074
GGA+U	Co <sub>2</sub> VIn	2.031	1.091	-0.074	-0.078
	CoVIn	0.902	-0.532	1.370	0.067
PBE-GGA	Co <sub>2</sub> VIn	2.011	0.968	0.131	-0.056
	CoVIn	0.879	-0.020	0.829	0.070
PBE-GGA+U	Co <sub>2</sub> VIn	2.030	1.069	-0.029	-0.078
	CoVIn	0.919	-0.780	1.658	0.041

(FLAPW) within PBE- GGA.  $^{18}$  This trend is well replicated in our results where a reduction in the V moment is resulting in an increase in the Co moment, with the total moment varying not by more than  $0.04\mu B$ . The total magnetic moment of  $Co_2VIn$  is mainly contributed by the Co and V sites, where these contributions are due to the large exchange splitting in the Co and V atoms for the majority-spin and minority-spin channels as depicted by the partial density of states plots in Figure 1.

The In atoms have a negligible local moment with opposite sign in comparison with the Co and V elements. In this regard, In, though larger than Al and Ga yields a similar outcome. Its 5p states have an anti-parallel interaction due to p-d hybridization to the 3d Co and 3d V orbitals of the transition elements.

All except GGA and L(S)DA schemes predict a ferrimagnetic coupling between Co and V in CoVIn. In a study of magnetic properties of  $Co_2FeSi$ , L(S)DA+U was found to yield magnetic moment values in agreement with experiment but not GGA or LDA. <sup>19</sup> In a study of the electronic properties of CoVSb, the moments of Co and V were found to be -0.20 and 1.12  $\mu$ B using PBE-GGA. <sup>18</sup>

As is evident from Table II, the more electron correlation is put into consideration, the higher the total magnetic moments. In the case of CoVIn, 3d V states contribute most of the moment.

# IV. CONCLUSION

While Co<sub>2</sub>VIn displays conventional half metallic characteristics, CoVIn displays a half metallic gap in the majority states, with neither the application of an external field nor doping. Therefore, these findings highlight two new promising half-metallic materials toward realistic spintronics applications.

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<sup>&</sup>lt;sup>1</sup> T. Graf, C. Felser, and S. S. P. Parkin, "Simple rules for the understanding of Heusler compounds," Prog. Solid State Chem. **39**(1), 1–50 (2011).

<sup>&</sup>lt;sup>2</sup> B. K. Hazra, M. M. Raja, and S. Srinath, "Correlation between structural, magnetic and transport properties of Co 2 FeSi thin films," J. Phys. Appl. Phys. **49**(6), 65007 (2016).

<sup>&</sup>lt;sup>3</sup> I. Galanakis, K. Özdoğan, E. Şaşıoğlu, and B. Aktaş, "Ferrimagnetism and antiferro-magnetism in half-metallic Heusler alloys," Phys. Status Solidi A 205(5), 1036–1039 (2008).

<sup>&</sup>lt;sup>4</sup>L. Feng, E. K. Liu, W. X. Zhang, W. H. Wang, and G. H. Wu, "First-principles investigation of half-metallic ferromagnetism of half-Heusler compounds XYZ," J. Magn. Magn. Mater. **351**, 92–97 (2014).

<sup>&</sup>lt;sup>5</sup> R. Umamaheswari, M. Yogeswari, and G. Kalpana, "Ab-initio investigation of half-metallic ferromagnetism in half-Heusler compounds XYZ (X=Li, Na, K and Rb; Y=Mg, Ca, Sr and Ba; Z=B, Al and Ga)," J. Magn. Magn. Mater. **350**, 167–173 (2014).

<sup>&</sup>lt;sup>6</sup> A. W. Carbonari, R. N. Saxena, W. Pendl, Jr., J. Mestnik Filho, R. N. Attili, M. Olzon-Dionysio, and S. D. de Souza, "Magnetic hyperfine field in the Heusler alloys Co2YZ (Y = V, Nb, Ta, Cr; Z = Al, Ga)," J. Magn. Magn. Mater. **163**(3), 313–321 (1996).

- <sup>7</sup> S. Trudel, O. Gaier, J. Hamrle, and B. Hillebrands, "Magnetic anisotropy, exchange and damping in cobalt-based full-Heusler compounds: an experimental review," J. Phys. Appl. Phys. 43(19), 193001 (2010).
- <sup>8</sup> W. Wang, H. Sukegawa, and K. Inomata, "Perpendicular magnetic anisotropy of Co<sub>2</sub> FeAl/Pt multilayers for spintronic devices," Appl. Phys. Express 3(9), 93002 (2010).
- 9 M. A. Tanaka, Y. Ishikawa, Y. Wada, S. Hori, A. Murata, S. Horii, Y. Yamanishi, K. Mibu, K. Kondou, T. Ono, and S. Kasai, "Preparation of Co2FeSn Heusler alloy films and magnetoresistance of Fe/MgO/Co2FeSn magnetic tunnel junctions," J. Appl. Phys. 111(5), 53902 (2012).
- <sup>10</sup> L. Wollmann, S. Chadov, J. Kübler, and C. Felser, "Magnetism in cubic manganese-rich Heusler compounds," Phys. Rev. **B 90**, 214420 (2014).
- 11 Z. Wen, T. Kubota, T. Yamamoto, and K. Takanashi, "Fully epitaxial C1b-type NiMnSb half-Heusler alloy films for current-perpendicular-to-plane giant magnetoresistance devices with a Ag spacer," Sci. Rep. 5, 18387 (2015).
- 12 X.-P. Wei, J.-B. Deng, G.-Y. Mao, S.-B. Chu, and X.-R. Hu, "Half-metallic properties for the Ti2YZ (Y=Fe,Co,Ni,Z=Al,Ga,In) Hesuler alloys: A first-principles study," ArXiv11105411 Cond-Mat, Oct. 2011.
- <sup>13</sup> Z. Ren, Y. Liu, S. Li, X. Zhang, and H. Liu, "Site preference and electronic structure of Mn2RhZ (Z = Al, Ga, In, Si, Ge, Sn, Sb): a theoretical study," Mater. Sci.-Pol. **0**(0) (2016).
- <sup>14</sup> S. N. Holmes and M. Pepper, "Cobalt-based Heusler Alloys for spin-injection devices," J. Supercond. **16**(1), 191–194
- <sup>15</sup> P. Klaer, "Disentangling the Mn moments on different sublattices in the half-metallic ferrimagnet Mn3? xCoxGa," Appl. Phys. Lett. (2012).
- <sup>16</sup> Y. Ma, X. Li, Q. Sun, and B. Huang, "Prediction of two-dimensional materials with half-metallic dirac cones: Ni2C18H12 and Co2C18H12." Elsevier 73, 382–388 (2014).
- <sup>17</sup> J. C. Woo, H. Hasegawa, Y. S. Kwon, T. Yao, and K. H. Yoo, Compound Semiconductors 2004: Compound Semiconductors for Quantum Science and Nanostructures (CRC Press, 2005).
- <sup>18</sup> S. E. Kulkova, S. V. Eremeev, T. Kakeshita, S. S. Kulkov, and G. E. Rudenski, "The electronic structure and magnetic properties of full- and half-Heusler alloys," Mater. Trans. 47(3), 599-606 (2006).
- <sup>19</sup> H. C. Kandpal, G. H. Fecher, C. Felser, and G. Schönhense, "Correlation in the transition-metal-based Heusler compounds Co<sub>2</sub>MnSi and Co<sub>2</sub>FeSi," Phys. Rev. B 73(9), 94422 (2006).
- <sup>20</sup> M. ElAmineMonir, R. Khenata, H. Baltache, G. Murtaza, M. Abu-Jafar, D. Rached, S. BinOmran, and A. Bouhemadou, "Study of structural, electronic and magnetic properties of CoFeIn and Co2FeIn Heusler alloys," jmmm, Journal of Magnetism and Magnetic Materials 394, 404-409 (2015).
- <sup>21</sup> T. Graf, C. Felser, and S. S. P. Parkin, "Heusler compounds: Applications in spintronics," in *Handbook of Spintronics*, edited by Y. Xu, D. D. Awschalom, and J. Nitta (Springer, Netherlands, 2016), pp. 335-364.
- <sup>22</sup> H. C. Kandpal, G. H. Fecher, and C. Felser, "Calculated electronic and magnetic properties of the half-metallic, transition metal based Heusler compounds," J. Phys. Appl. Phys. 40(6), 1507 (2007).
- <sup>23</sup> S. Krishnaveni, M. Sundareswari, and M. Rajagopalan, "Prediction of electronic and magnetic properties of full Heusler Alloy-Ir2CrAl," IOSR J. Appl. Phys. IOSR-JAP 7, 52-55 (2015).